DEVELOPMENT OF HVOF SPRAYED EROSION/OXIDATION RESISTANT COATINGS FOR COMPOSITE STRUCTURAL COMPONENTS IN PROPULSION SYSTEMS

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Thermally sprayed coatings are being studied and developed as methods of enabling lightweight composites to be used more extensively as structural components in propulsion applications in order to reduce costs and improve efficiency through weight reductions.

The primary goal of this work is the development of functionally graded material [FGM] polymer/metal matrix composite coatings to provide improved erosion/oxidation resistance to polyimide-based polymer matrix composite [PMC] substrates. The goal is to grade the coating composition from pure polyimide, similar to the PMC substrate matrix on one side, to 100 % WC-Co on the other. Both step-wise and continuous gradation of the loading of the WC-Co reinforcing phase are being investigated.

Details of the coating parameter development will be presented, specifically the high velocity oxy-fuel [HVOF] combustion spraying of pure PMR-II matrix material and layers of various composition PMR-II/WC-Co blends onto steel and PMR-15 composite substrates. Results of the HVOF process optimization, microstructural characterization, and analysis will be presented. The sprayed coatings were evaluated using standard metallographic techniques - optical and scanning electron microscopy [SEM]. An SEM + electron dispersive spectroscopy [EDS] technique has also been used to confirm retention of the PMR-II component. Results of peel/butt adhesion testing to determine adhesion will also be presented.

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Primary Goal:

composite [PMC] substrates for use in turbine engine components erosion/oxidation resistance to lightweight polyimide-based polymer matrix Development of functionally graded material [FGM] coatings to provide

Secondary Goals:

- Improved surface finish relative to coatings developed in a previous study
- Some evaluation of the thermo-mechanical fatigue [TMF] properties of the coating/substrate system.

Objectives:

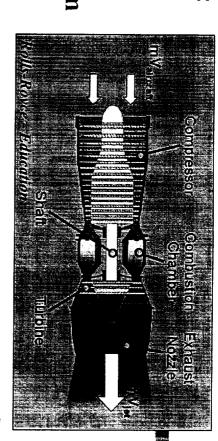
Development of thermally sprayed FGM coating structures based on a polyimide matrix filled with varying volume fractions of conventional or nano-sized WC-Co.

- Advantages of PMC's in Aircraft Engines:
- Weight Savings
- Improved Strength
- Reduced Part Count

Reduced Mfg. Cost

propulsion system

More efficient

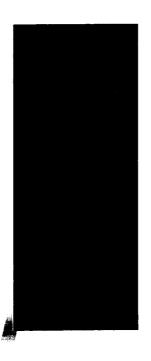


■ New Generation PMC...PMR-II: Erosion-Oxidative Resistant Coating Long Term Operation up to ~370 °C

Increased # of Potential Applications

LOW COST, HIGH TEMPERATURE POLYMER MATRIX COMPOSITE [PMC] COMPONENTS:

- Allison Advanced Development Company, Indianapolis, IN.
- Rolls-Royce Allison, Indianapolis, IN.
- Ohio Aerospace Institute, Cleveland, OH.
- Real World Quality Systems, Rocky River, OH.



I Statistical Analysis of Six Different Erosion Resistant Coating Systems:

TiN	WC-Co	Cr ₃ C ₂ - NiCr	TiN	WC-Co		Cr ₁ C ₂ - NiCr	Top Coat Composition F
В	В	В	A	\boldsymbol{A}		A	Bond Coat Type
OMC	ОМС	OMC	OMC	ОМС	Composite [OMC]	Organic Matrix	Substrate

Key Conclusions:

- WC-Co [A] exhibited the best overall performance.
- Provided on average 8.5 times greater improvement in erosion resistance as compared with uncoated PMR-15 [OMC].

Development & understanding of the thermal spray processing techniques needed to produce these.

Characterization coatings: Of: thermally sprayed FGM

Microstructure.

Matrix degradation.

Surface roughness.

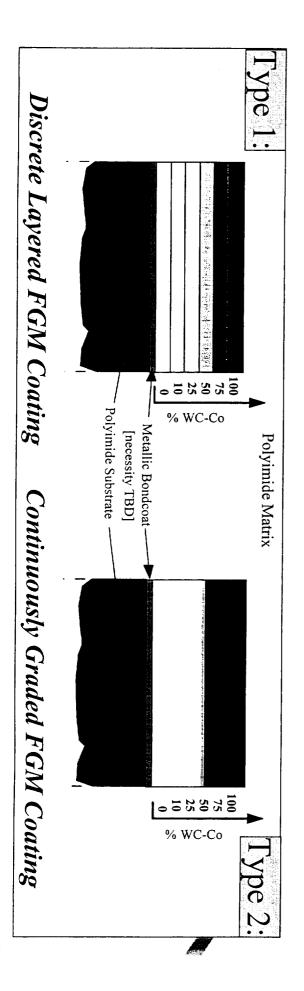
Determination properties. behavior and of the erosion-resistance, oxidation thermo-mechanical fatigue [TMF]

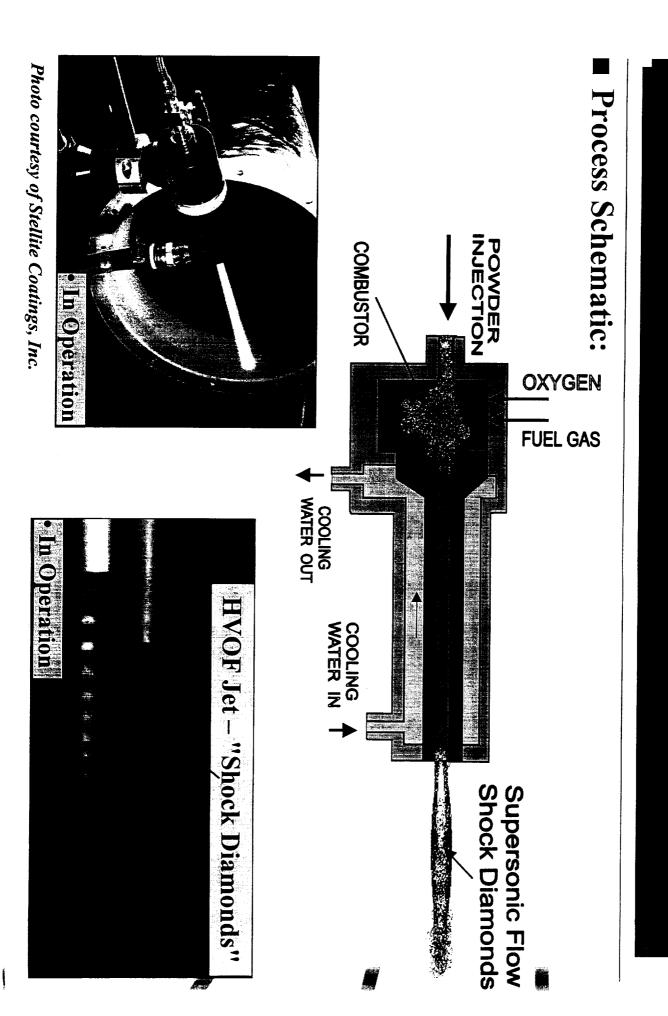
■ Matrix: *Polyimide PMR-II*

and PMC substrate match differences in thermal expansion coefficient [CTE] between WC-Co Improve the TMF properties of the coating/substrate system by helping

■ Reinforcement: *WC-Co*.

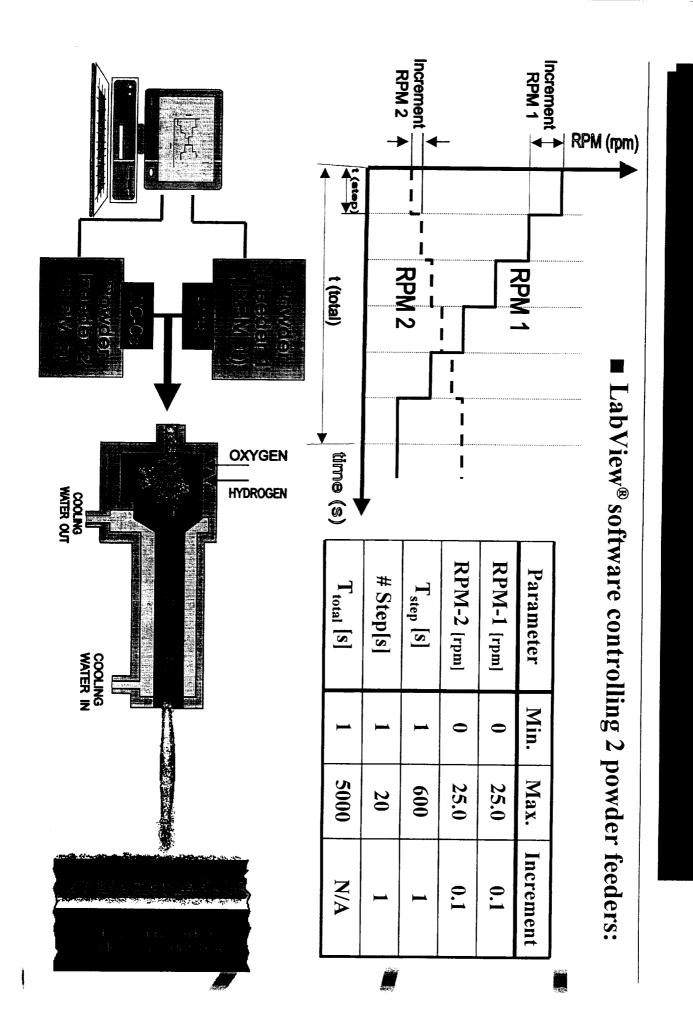
Erosion/oxidation-resistant reinforcing phase and outer layer.





Process History/Characteristics:

- Invented c. 1958 by Union Carbide [now Praxair].
- Introduced commercially in 1974 [J. Browning]; now...5-8 designs [Jet Kote®, Diamond-Jet, HV-2000, JP-5000, OSU, etc.].
- Carbides [WC/Co, Cr₃C₂/NiCr], metals [SS]...aircraft engines, industry.
- Pressurized internal combustion + supersonic jet expansion to atmosphere
- Fuel [H₂, C₃H₆, C₃H₈, MAPP, Kerosene] + Oxygen @ high pressure
- Improved gas heating and acceleration...improved heat/momentum transfer to injected powder particles
- Powder feedstocks [typ. $(-45, +10 \mu m)$].
- Characteristics:
- Jet Temperature: typ. >3500 °C.
- Jet Speeds: typ. >1200 m/s.
- Gas Flow Rates: 400-1100 slm (a) 100-140 psi.
- Particle Speeds: 200-1000 m/s.
- Powder Feed Rates: 15-50 gm/min.



Feedstock Materials:

- PMR-II [n=2; n=9] [ρ = 1.39 g/cm³] Pre-imidized molding powder.
- Cryo-ground to particle size distribution in range \sim [-100, + 20 μ m].

- WC-Co [ρ = 12.5 g/cm³]
 Agglomerated & sintered powder: [-44, + 10 μm].
 H.C. Starck: Amperit 518; Praxair: WC 106; Stellite Coatings, Inc: JK

Coating Production:

- Stellite Coatings, Inc. Jet Kote® II-A HVOF combustion spray.
- Praxair model 1270 volumetric powder feeders [2].

Characterization & Testing:

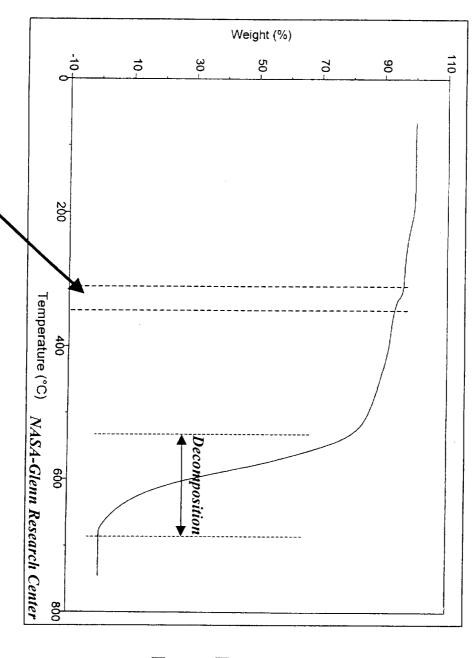
- Microstructure: Metallography; Optical Microscopy; SEM/EDS.
- Adhesion: Butt adhesion; 180° peel test; dynamic shear tests
- Stylus-tracing profilometry.
- Matrix Degradation: FTIR; XPS; Dilute-solution viscometry.
- Thermo Mechanical Fatigue [TMF]: MTS high temp. mechanical testing.

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- I Thermosetting polyimide [end-capped].
- Oligomer.
- Imidized ~2 hr. @ 400 °F + 1 hr. @ 450 °F.
- Mechanical crushing --> angular/blocky particle morphology.
- Screening --> particle size distribution for spraying.

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Thermo-Gravimetric Analysis [TGA]:



PMR-II [n=2]TGA Conditions:

8.576 mg

1:1 B1:B2

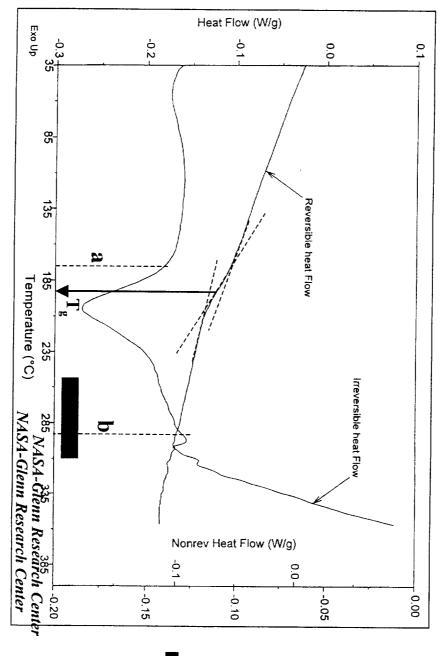
Universal V2.5H TA Instruments Run in Air

- High temp. properties of PMR-II polyimide.
- Degradation begins at ~550 °C.

wt. % at temp. ~ 325 °C. --> porosity in the coating.

Gas evolution [eg. water vapor] during heating resulted in weight loss of $\sim 3-5$

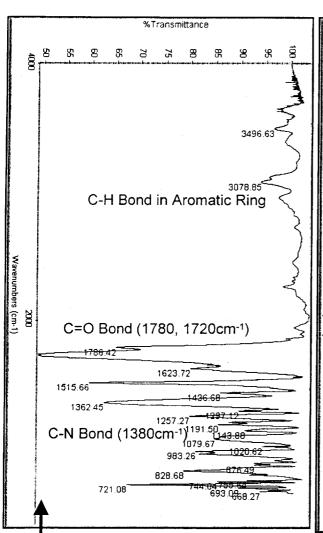
Differential Scanning Calorimetry [DSC]:

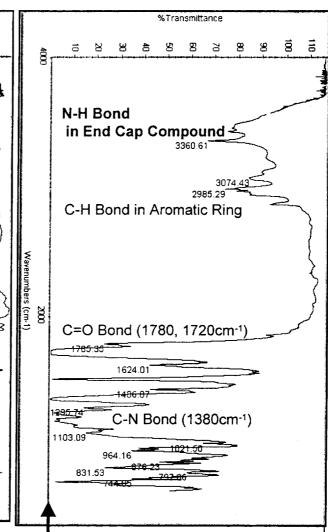


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■ Reversible Heat Flow: $T_g \sim 190$ °C of PMR-II $\frac{before}{}$ end groups are polymerized.

I **Irreversible Heat Flow**: [a] Melting or fusion of oligomer began [~170 °C]. [b] Onset of polymerization of end-groups [~300 °C]. ...Substrate pre-heating region.





Infrared] Spectrometry:

- Nicolet Magna-IR560
- spectrometer.
 Changes in molecular
- structure of PMR-II powders before and after heating.
 -Received" PMR-II [n = 2] Powde

"As-Received" PMR-II [n = 2] Powder.

Polymerization after heat-treating indicated by disappearance of N-H bond peak characteristic of the powder, and development of C-N bond peaks [with the N bonded to C atoms of aromatic rings] in range 1030 to 1230 cm⁻¹.

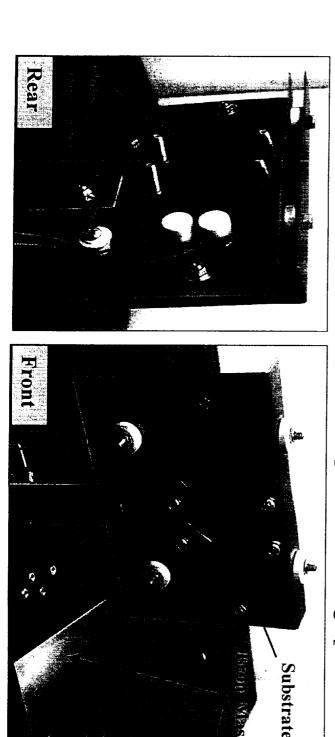
PMR-II [n = 2] Powder Heated @ 350

SSues

- Rapid heating & cooling characteristic of HVOF spray process.
- High heat flux from the process/coating into the substrate
- Insufficient time @ temperature for completion of curing of the thermosetting polymer.
- Initial results indicated T_{substrate} to be a key parameter influencing deposition & coating build-up

Proposed Solution:

- Electrical resistance driven substrate preheating.
- Substrates can be preheated to temperatures in range [280-320 °C].



Side

Vee-Blending: [10 min]

effectiveness of subsequent ball-milling. To homogenize the materials and improve





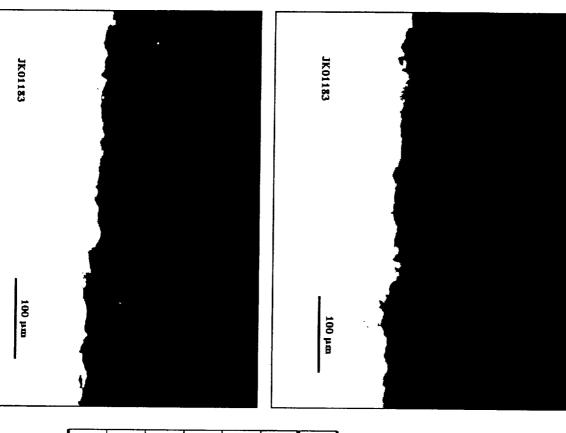
Milling

Ball-Milling: [30 min @ 55 rpm]

50/50 Vol. % PMR-II & WC-Co materials were used throughout the ball-milling process. Equal nominal volumes [~450 ml] ZrO₂ balls +



- PMR-II particle size reduced from initial relatively coarse [-150, +102 µm] to approx. the same size as the WC-Co [$\sim 50 \mu m$] during ball-milling.
- observed Little or no embedding of the hard WC-Co particles into the softer PMR-II particles



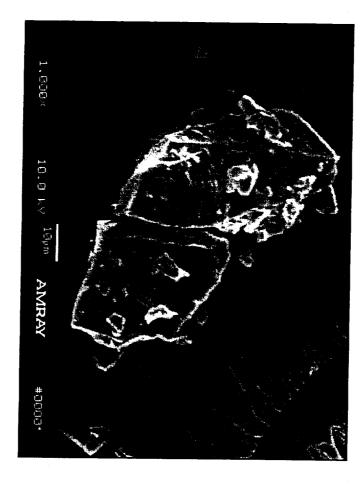
- HVOF sprayed, pure PMR-II coatings on Steel substrates [Run # JK01183].
- Same sample...different location [200X].
- · Relatively dense, adherent coatings.
- Some porosity…likely due to gas/moisture evolution.

HVOF Process ParameterValueSpray Distance [in]6Powder Feed Rate [rpm]1.5Gas Flow Rates; H2:O2 [SCFH]800:400Nozzle6" L x 0.25" ØSurface Speed [in/s]4.5 [50 %]*		280-320 [1.7-1.8 A]**	Substrate Temperature [°C]
		4.5 [50 %]*	Surface Speed [in/s]
	hour	6" L x 0.25" Ø	Nozzle
		800:400	Gas Flow Rates; H ₂ :O ₂ [SCFH]
		1.5	Powder Feed Rate [rpm]
		9	Spray Distance [in]
	and the same	Value	HVOF Process Parameter

^{*} Max. horizontal gun [surface] speed = 9 in/s [45 ft./min.].

^{** 1.7-1.8} A = current [A] corresponding to the indicated substrate temperature.

SEM/EDS: "dot-mapping" analysis of PMR-II powder.



SEM Image [1,000X]



Fluorine "dot-map"

Fluorine dot-map clearly corresponded to the morphology of the particle[s] presence of the PMR-II material. --> confirmed viability of proposed approach for detecting/confirming the

SEM/EDS dot maps of HVOF sprayed pure PMR-II coating: [Run #JK01161c; Al Substrate]



Secondary Electron Image



Aluminum



Fluorine

Carbon

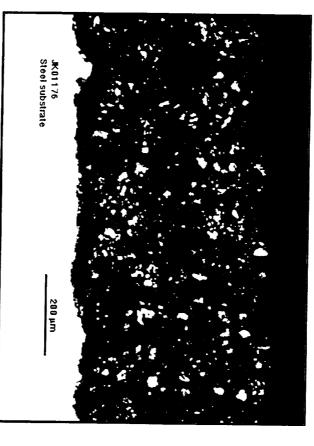
Results confirmed sprayed PMR-II coatings. characterizing HVOF suitability of SEM/EDS technique for

Optical Micrograph: HVOF sprayed pure PMR-II coating on PMR-15 PMC substrate [100X].

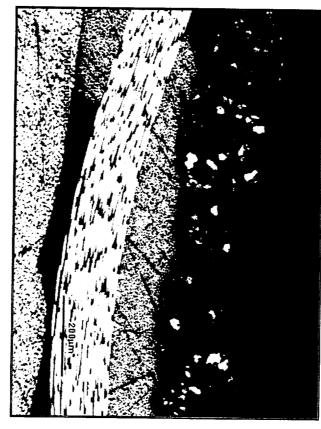


- Good coating/substrate interface.
- Relatively low porosity.

Optical Micrographs: HVOF sprayed 90 % PMR-II/10 % WC-Co composite coatings:



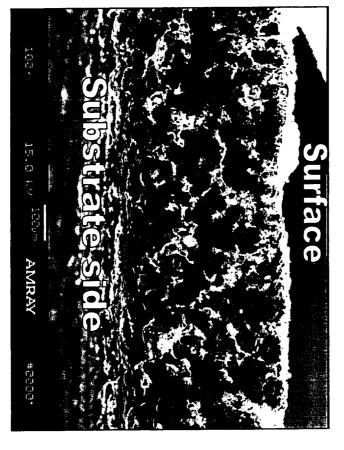
Steel Substrate [200 X]



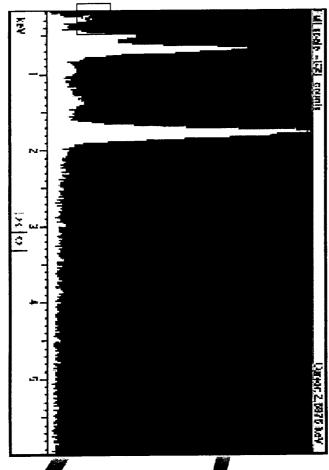
PMR-15 Substrate [200X]

- Good coating/substrate interface.
- Uniform dispersion of WC-Co phase.
- Increased porosity on PMR-15 substrate [likely due to water vapor/gas evolution].

SEM micrograph & corresponding EDS profile of a fractured cross-section substrate. from a 90 Vol. % PMR-II/10 Vol. % WC-Co coating, removed from



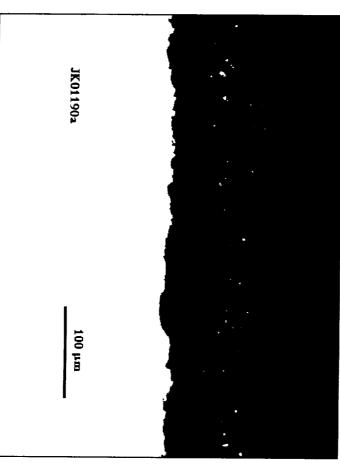
SEM Micrograph [100X]



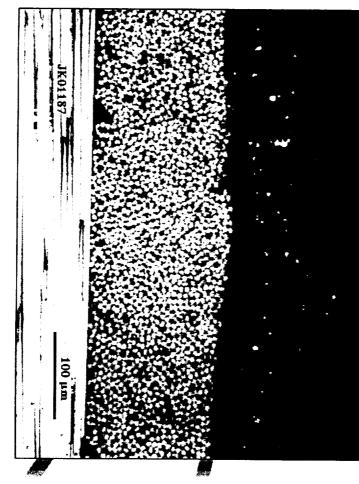
EDS Spectrum

Successful deposition & retention of both PMR-II and WC-Co components of Vee-blended feedstock in sprayed deposit

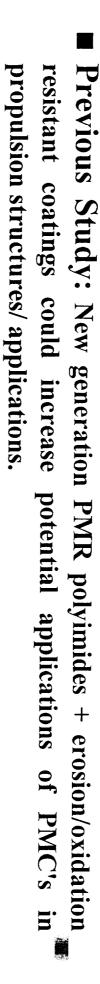
- HVOF Sprayed, 2-layer Coating on Steel Substrate: [200X]:
- 1st layer...pure PMR-II.
- 2nd [top] layer...50:50 Vol. % PMR-II/WC-Co.



■ HVOF Sprayed 90:10 PMR-II/WC-Co Coating on PMR-15 Substrate.



- Good coating/substrate interface.
- Two distinct layers clearly visible.
- WC-Co content does not appear to be 50 %...Ashing in-progress.



Current Program:

- Feasibility of depositing PMR-II and PMR-II + WC/Co demonstrated.
- Dense, adherent coatings HVOF sprayed onto steel, Al, PMR-15.
- SEM/EDS technique [F] established for confirming retention of PMR-II in HVOF sprayed PMR-II coatings.
- FTIR spectroscopy found to be suitable for characterizing molecular structure changes during thermal processing of PMR-II polyimide.
- Dual layer [Type 1] FGM PMR-II/WC-Co coatings successfully
- Initial results indicate that T_{substrate} = key parameter influencing deposition & coating build-up, due to rapid heating & cooling characteristic of HVOF spray process.



- Continued spray parameter optimization and assessment of repeatability.
- Measurement of thermoset crosslinking density via percent mass swelling.
- Measurement of coating adhesion via tensile and/or peel type adhesion tests.
- stylus-tracing profilometry. Measurement of "as-sprayed" surface roughness [Ra] of FGM coatings by
- Erosion testing of Type 1 FGM coatings.
- layer of FGM coatings by FTIR and EDS, respectively. Determination of oxidation resistance of polyimide matrix & WC-Co outer
- using high temperature MTS mechanical testing system. Study thermo-mechanical fatigue [TMF] behavior of coating/substrate system